



Influence of annealing on Si-Ge thin films with high concentration of tellurium and zinc

¹Dr.Ibrahim A. Saleh.

1. Assistant Professor of Physics Department, Faculty of Arts and Science "Al Abyar", Benghazi University, Abraheem.salih@uob.edu.ly.

DOI: <https://doi.org/10.37376/ajhas.vi2.6862> Received: 11.05.2024 Accepted: 20.05.2024 Published: 16.08.2024

Abstract:

The elements are distinguished via the basis of energy band gap. Two configurations $\text{Si}_{10}\text{Ge}_{10}\text{Te}_{80}$ and $\text{Si}_{10}\text{Ge}_{10}\text{Zn}_{80}$ multilayers were annealed under vacuum at eutectic temperature of 400°C to form alloys between the deposited layers. The two samples were studied at different temperatures to improve the physical properties of the thin films. Kinetics of the two configurations $\text{Si}_{10}\text{Ge}_{10}\text{Te}_{80}$ and $\text{Si}_{10}\text{Ge}_{10}\text{Zn}_{80}$ were studied in the temperature range $308\text{--}673^{\circ}\text{K}$. The results revealed that the crystallization of the thin film of multilayer for the samples increases with increasing the annealing temperature and increasing the zinc content. The influence of annealing on the structure and stability of the samples has been studied by X-ray diffraction. The optical and electrical measurements were carried out after annealing. The electrical conductivity of the two samples was measured as a function of temperature and annealing time. It was found that the electrical conductivity for $\text{Si}_{10}\text{Ge}_{10}\text{Zn}_{80}$ increases, and the optical gap decrease due to the crystallization effects that occurs only in Ge matrix and with increasing zinc content at high concentration.

Keywords: thin films, annealing, temperature, crystallization, optical gap, electrical conductivity.

Copyright © C2024 University of Benghazi.

This open Access article is Distributed under a [CC BY-NC-ND 4.0 license](#)



الرقيقة ذات التركيز العالي من التيلوريوم والزنك (Si-Ge) تأثير التلدين على أغشية

د. إبراهيم صالح عبد الحفيظ.^١

١. أستاذ مساعد بقسم الفيزياء، كلية الآداب والعلوم "الأبيان"، جامعة بنغازي.

الملخص

يتم تمييز العناصر من خلال فجوة نطاق الطاقة الأساسية. تم التلدين بتكتوينين $\text{Si}_{10}\text{Ge}_{10}\text{Te}_{80}$ و $\text{Si}_{10}\text{Ge}_{10}\text{Zn}_{80}$ متعدد الطبقات تحت فراغ عند درجة حرارة سهلة الانصهار تبلغ 400 درجة مئوية لتشكيل سبائك بين الطبقات المودعة. تمت دراسة العينتين عند درجات حرارة مختلفة لتحسين الخواص الفيزيائية للأغشية الرقيقة. تمت دراسة حركية التكتوينين $\text{Si}_{10}\text{Ge}_{10}\text{Zn}_{80}$ و $\text{Si}_{10}\text{Ge}_{10}\text{Te}_{80}$ في المدى الحراري 308-673 درجة مئوية. أظهرت النتائج أن تبلور الطبقات الرقيقة للعينات يزداد مع زيادة درجة حرارة التلدين وزيادة محتوى الزنك. تمت دراسة تأثير التلدين على بنية وثبات العينات باستخدام حبيبات الأشعة السينية. تم إجراء القياسات الضوئية والكهربائية بعد التلدين. تم قياس التوصيل الكهربائي للعينتين كدالة لدرجة الحرارة وזמן التلدين. وجد أن الموصولة الكهربائية لـ $\text{Si}_{10}\text{Ge}_{10}\text{Zn}_{80}$ تزداد، وتقل الفجوة الضوئية بسبب تأثيرات التبلور التي تحدث فقط في مصفوفة Ge مع زيادة محتوى الزنك عند التركيز العالي.

الكلمات المفتاحية: الأغشية الرقيقة، التلدين، درجة الحرارة، التبلور، الفجوة الضوئية، التوصيل الكهربائي.

Scan QR & Read Article Online.





1. Introduction

The modern technological era started with the invention of the transistor and the birth of solid state electronics. From that date, we have been rapidly developed the science and technology based firstly on germanium, then on silicon, and later on multilayers for elements with germanium, and silicon. The necessity to save weight and ratio of element, and to improve the electrical conductivity in the function of the devices, stimulated an intense research activity to obtain the thin films from multilayers. In the same time, the worldwide interest was moved from electronic to more complex optoelectronic applications. Taking into account these big tasks, the class of materials must be enlarged, new materials with new properties must be prepared. The physical properties of thin films materials of many semiconductors and elements has been investigated. [18].

The improvement of devices based on the prepared materials via different high technology fields of science. Nowadays, commercial devices require the development of tailored materials. Approaches to evaluate the phy-

sical properties of their structures produces an inevitable industrial delay in the devices improvement. Moreover, with the trend to continuous curing of the components makes it becomes necessary to deal with the spatial resolution problem. For the advance in such applications, the locally investigation of the change in the electrical conductivity with temperature and elements contents of in new compounds of materials is therefore a key issue.

Materials are commonly affected by a large concentration of point defects, hence the heat treatment are very important. The transition of crystalline state proceeds by nucleation and growth reactions, a systematic study on the kinetics of the crystal phase remains one of the most interesting aspects of physics properties. [7].

A strict control on the ratio of materials and location is necessary, but also because their behavior in the material structure is still rather unclear. Electrical conductivity may also be used to investigate of crystallization for material of thin films and type of elements. [9].

The aim of this work is to study the physical properties of the thin film



s.for.multilayers.and.using.the.results.t
o.investigate.affects.on.the.electrical.co
nductivity.with.annealing.and.high.con
centration.of.materials.contents.

II.Experimental

Two.configurations.of.multila
yers.were.deposited.using.Edwards.3
06.thermal.evaporator.under.vacuum.
of. 6×10^5 .mbar.on.borosilicate.glass.s
ubstrates..The.first.configuration.con
sists.of.silicon(Si),.germanium.(Ge),.
and.tellurium.(Te),.and.the.second.co
nfiguration.consists.of.silicon.(Si),.g
ermanium.(Ge),.and.Zinc.(Zn)..The.
Two.samples.prepared.at.80°C...The.
first.sample.in.this.configuration.is.S
 $_{10}$ Ge $_{10}$ Te $_{80}$.and.the.thicknesses.of.lay
ers.of.this.sample.are.30.nm.of.Si,.30
.nm.of.Ge,.and.240.nm.of.Te..The.se
cond.sample.in.this.configuration.is.
Si $_{10}$ Ge $_{10}$ Zn $_{80}$.and.the.thicknesses.of.1
ayers.of.this.sample.are.30.nm.of.Si,.
30.nm.of.Ge,.and.240.nm.of.Zn..The.
samples.of.the.two.configurations.are
.then.annealed.under.vacuum.at.eute
ctic.temperature.of.400.°C.to.form.al
loys.between.the.deposited.layers..T
he.following.table.shows.the.prepare
d.samples..The.total.film.thickness.m

easured.by.the.Dectak.surface.profile
r.was.in.the.range.of.300nm.
The.cocrystallinity.of.the.two.samples.
was.measured.by.Xray.diffractomete
r.(Model:.Labx.XRD6100.).provide
d.by.copper.(Cu).Xray.tube.of.wavel
ength. $\lambda=1.54\text{\AA}$.was.used.for.structure
analysis.of.the.thin.films..The.interpl
aner.distance.was.determined.using.
Bragg's.Equation: $2d.\sin\theta=n\lambda$
Where.d.is.the.interplaner.distance.a
nd. θ .is.the.angle.of.x-ray.diffraction.
JASCO.V630.UVVIS.Spectrophotome
ter.was.used.for.transmission.and.
absorption..measurements..A.cryosta
t.(.photon.cryostat.220.Ohmmetry.).
was.used.for.electrical.measurements
The.sample.is.maintained.between.t
wo.electrodes.as.a.coplanar.structure.
attached.to.a.dc.voltage.of.60.Volt..T
he.pressure.is.about. 6×10^3 .mbar,.dep
ending.on.the.geometry.of.the.vacuu
m.chamber..The.substrate.holder.is.p
rovided.by.heating.system.to.raise.th
e.temperature.of.the.sample.up.to.60
°C.



III. Results and discussion

1. The X-ray diffraction (XRD)

The structural properties of $\text{Si}_{10}\text{Ge}_{10}\text{Te}_{80}$ and $\text{Si}_{10}\text{Ge}_{10}\text{Zn}_{80}$ prepared at 80°C and annealed at 400°C have been investigated by XRD. The phases appeared in the x-ray spectra are GeGe(111), GeGe(2

Table(1): The d-spacing of $\text{Si}_{10}\text{Ge}_{10}\text{Te}_{80}$ and $\text{Si}_{10}\text{Ge}_{10}\text{Zn}_{80}$ prepared at 80°C after annealing at 400°C for 2h.

Phases	(2θ) . $^{\circ}$ 2Theta.	d-Spacing.(\AA)	$h^2+k^2+l^2$	Orientations	Lattice constant.(\AA)	Lattice Constant.(\AA).in.text.book.[10-11].
Ge(111)	26.889	3.311	3	(111)	5.736	5.74
Ge(211)	38.626	2.328	6	(211)	5.702	5.6575.and.5.74
Ge(220)	44.953	2.014	8	(220)	5.700	5.6575.and.5.74

It is clear from x-ray analysis of the two samples that crystallization effects occur only in the Ge matrix because the crystallizing temperature of the germanium (about 300°C) is lower than that of silicon (about 500°C). This also indicates that zinc atoms are located between crystalline multilayers, that atoms also conformational bonds with the surrounding atoms..

Therefore, it has a good bond and reduces crystal defects and increases electrical conductivity, where the crystallization effects occur in the sample $\text{Si}_{10}\text{Ge}_{10}\text{Zn}_{80}$ more than the crystallization occurs in the sample $\text{Si}_{10}\text{Ge}_{10}\text{Te}_{80}$.

11), GeGe(220) for sample $\text{Si}_{10}\text{Ge}_{10}\text{Zn}_{80}$ and phases appeared in the x-ray spectra are GeGe(111), GeGe(211) for sample $\text{Si}_{10}\text{Ge}_{10}\text{Te}_{80}$. The inter-plane distances (dspacing) have been calculated using the Bragg's equation and are given in table (1)..

2. Optical Properties

In this section we will discuss the measurements of transmission of $\text{Si}_{10}\text{Ge}_{10}\text{Te}_{80}$ and $\text{Si}_{10}\text{Ge}_{10}\text{Zn}_{80}$ thin films prepared at 80°C , and the deduced optical energy gap after annealing at 400°C .

2.1. Transmission and absorption measurements.

Figures (1) show transmission for the $\text{Si}_{10}\text{Ge}_{10}\text{Te}_{80}$ and $\text{Si}_{10}\text{Ge}_{10}\text{Zn}_{80}$ thin films thin films after annealing at 400°C . It is clear that for high energies (λ_{\min}) there is no transmission appeared for samples but for low energies (λ_{\max}) the transmission is high. For low energies (λ_{\max}) h



owever, there are no appropriate electronic transitions possible so transmission is very high in this range.. It is not 100%. however, because of absorption.. The interpretation of the transmission spectra in figure (1) is not so straight forward.. For high energies,, where absorption is high according to the transmission chart,, increases as well.. This can only be explained by a theoretical treatment of the interaction between light and matter.[12.1 4].

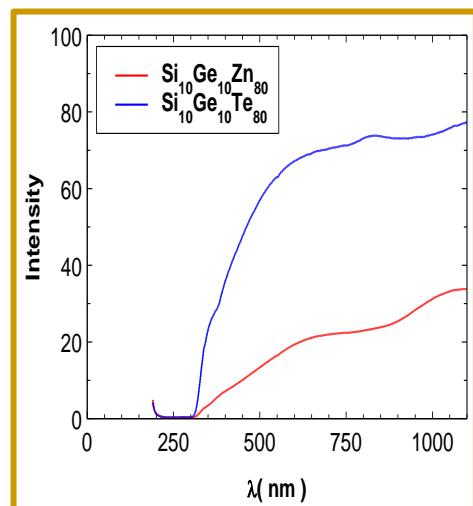


Figure.(1):.Transmission.spectra for untreated.the.Si₁₀Ge₁₀Te₈₀.and. Si₁₀Ge₁₀Zn₈₀.thin.films.prepared.at.8 0°C.and.annealed.at.400°C.for.2.h.It. is.shown.that.for.high.energies.(λ_{\min}). there.is.no.transmission.for.film.Si₁₀

Ge₁₀Te₈₀,but.for.film.Si₁₀Ge₁₀Zn₈₀.the transmission.is.about.34%..For.high.energies.(λ_{\min}),.where.absorption.is .high.according.to.the.transmission.chart..After.annealing.at.400°C,the.ab sorption.is.increases.due.to.the.increa sing.of.crystallization.and.the.high.c oncentration.of.zinc.(Zn).which.leads .to.a.decrease.of.the.transmission.int e nsity...

2.2..Optical.energy.data

The.optical.band.gap.is.useful. material.parameter..It.allows.to.compare.between.the.samples.Si₁₀Ge₁₀Te₈₀ and.Si₁₀Ge₁₀Zn₈₀.thin.films.based. materials.regarding.their.light.absorp tion.properties.[15.17]..The.concentr ation.of.tellurium.and.zinc.plays.an.i mportant.role.for.determining.electri cal.conductivity.and.the.optical.energ y.gap.for.the.samples.Si₁₀Ge₁₀Te₈₀.a nd.Si₁₀Ge₁₀Zn₈₀.,as.conformed.by.th e.results.obtained.in.previous.works[18]..Figures.(2)..shows.the.relation.b etween.($\alpha h\nu$)^{1/2}.and.(hv).for.the.two. samples..According.to.Tauc's.relatio n.[19],.the.optical.energy.gaps.were. deduced.from.the.plots.,.and.the.value s.are.given.in.tables.(2).for.the.two.s amples.after.annealing..

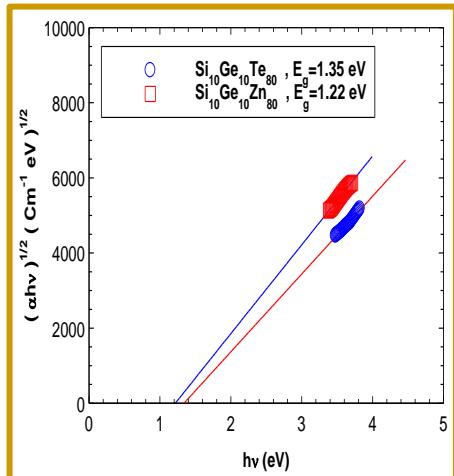


Figure.(2):.. $(\alpha h v)^{1/2}$.. Vs.. $h v$.for.
Si₁₀Ge₁₀Te₈₀.and.Si₁₀Ge₁₀Zn₈₀.thin.films.prepared.at.80°C.and.annealed.at.400°C.for.2.h.

The.data.shows.that.the.optical.energy.gap.is.decreases.with.raising.and/or.annealing.temperature.and.due.to.crystallization.effects.in.Ge.matrix.[20,23],.and.partially.due.to.high.concentration.of.zinc.(.Zn.).

Table.(2):.The.data.of.optical.band.gap.E_g.for.untreated.samples.

The.sample	E _g (eV.)
Si ₁₀ Ge ₁₀ Te ₈₀	1.35
Si ₁₀ Ge ₁₀ Zn ₈₀	1.22

3.Electrical.properties

3.1..Effects.of.temperature.and.(tellurium/zinc).high.concentration.on.the.electrical.conductivity.

The.conductivities.as.a.function.of.temperature.for.Si₁₀Ge₁₀Te₈₀.and.Si₁₀Ge₁₀Zn₈₀.thin.films.prepared.at.80°C.and.annealed.at.400°C.for.2.h..Here,.the.concentration.of.Te.and.Zn.are.80%,.in.the.temperature.range.308.473.K.are.shown.in.figures.(3).It.is.seen.that.the.relation.between.the.electrical.conductivity.and.the.temperature.obey.the.Arrhenius.type.equation: $\sigma=\sigma_0 \exp(E_a/kT)$.(1)

where. σ .is.electrical.conductivity., E_a .is.the.activation.energy.and. kB .is.the.Boltzmann`s.constant..The.high.concentration.for.elementsconducting.are.playing.an.important.role.for.determining.electrical.conductivity.of.multilayers.Si₁₀Ge₁₀Te₈₀.and.Si₁₀Ge₁₀Zn₈₀.thin.films..The.electrical.conductivity.measured.at.308.K.and.the.activation.energy.calculated.from.the.slopes.of.the.lines.for.the.samples.are.given.in.table.(3)..It.is.seen.that.the.electrical.conductivity.increases.while.activation.energy.decreases.with.increasing.the.content.of.(.Zn.).more.



than. (Te.) as conformed by the previous works. [18, 22, 23]..

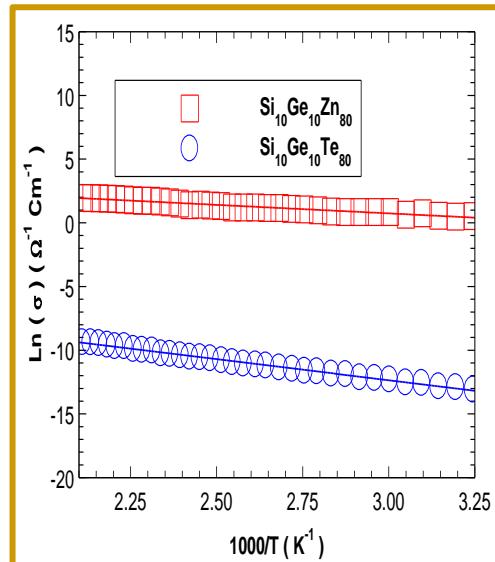


Figure.(3):.Dark.conductivity .vs..inverse.of.temperature.for.multilayers.of.Si₁₀Ge₁₀Te₈₀.and.Si₁₀Ge₁₀Zn₈₀.thin.films.

Table.(3):.Conductivity.measured.at.308.K.and.the.activation.energy.for.untreated.multilayers.of.Si₁₀Ge₁₀Te₈₀.and.Si₁₀Ge₁₀Zn₈₀.thin.films.

The.sample	concentration.(80%)	$\sigma(\Omega^{-1}.cm^{-1})$.	E _a (eV.)
Si ₁₀ Ge ₁₀ Te ₈₀	Te	26×10^{-3}	0.29
Si ₁₀ Ge ₁₀ Zn ₈₀	Zn	31×10^{-2}	0.12

3.2..Kinetics.and.thermal.stability

The.electrical.conductivity.of .multilayers.of.Si₁₀Ge₁₀Te₈₀.and.Si₁₀

Ge₁₀Zn₈₀.thin.films.as.a.function.of.a nnealing.times.were.recorded.at.different.temperatures.473,573.and.673.K .is.given..figure.(4).and.(5),.respectively..All.samples.show.the.same.general.behavior.where.the.electrical.conductivity.increases.with..increasing.the.annealing.time..at.constant.annealing..temperature.and.becomes.constant.at.high.annealing.time..This.behavior.is.attributed.partially.to.crystallization.occurring.in.the.multilayers.,that.crystallization.effects.occur..only.in.Ge.matrix.with.annealing..temperature..and..annealing.time..as..confirmed.by.Xray.data.,.and.also.depending.on.the.incorporation.of.zinc.or.tellurium.at.the.network.

It.is.known.that.the.conductivity.of.cristalline.material.is.higher.than.that.of.amorphous.one.since.the.ordered.systems.exhibit.lower.activation.energy.than.the.amorphous.one.[24].The.study.confirms.that.electrical.conductivity.increases.with.increases.at.high.concentration.of.the.element.doping..This.is.clearly.for.the.sample.doped.with.zinc.at.high.concentration,.more.than.for.the.sample.doped.with.tellurium..Thus..the.electrical.cond



uctivity..measurements..as.a.function ..of..annealing..time.at.constant..tem perature..are.used.to..study.the.isothe rmal.crystallization..kinetics.using.Jo hnson.Mehl.Avermi` s.(JMA).equatio n..

Thus.the.electrical.conductiv ty.measurements.as.a.function.of.ann ealing.time.at.constant.temperature.a re.used.to.study.the.isothermal.crysta llization.kinetics.using.Johnson- Mehl-.Avermi` s.(JMA).equation.in.t he.form[25]:

$$\chi = 1 - \exp[-kt^n] \quad (2)$$

Where. χ .is.the.volume.fractio n.of.the.crystalline.phases.transforme d.from.the.amorphous.state.at.time.t., n.refers.to.the.order.of.reaction.and.k .is.the.effective.overall.reaction.rate., which.actually.reflects.the.rate.of.cry stallization.[26,27].

The.electrical.conductivity.as .a.function.of.annealing.time.the.vol ume.fraction. χ .is.:

$$\chi = (\ln \sigma_a - \ln \sigma_t) / (\ln \sigma_a - \ln \sigma_c) \quad (3)$$

Where. $\ln \sigma_a$.is.logarithm.of.th e.electrical.conductivity.at.zero.time. (activation.electrical.conductivity),.L n. σ_t .logarithm.of.the.electrical.condu ctivity.at.any.time.t.and. $\ln \sigma_c$.is.logar

ithm.of.the.electrical.conductivity.at. the.end.of.saturation.(full.crystallizat ion).

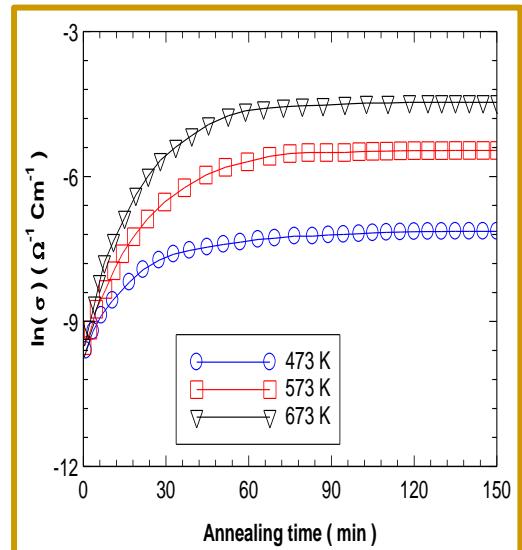


Figure.(4):.logarithm.of.the.el ectrical.conductivity.versus.the.anne aling.time.at.constant.temperatures.f or.Si₁₀Ge₁₀Te₈₀.multilayers.

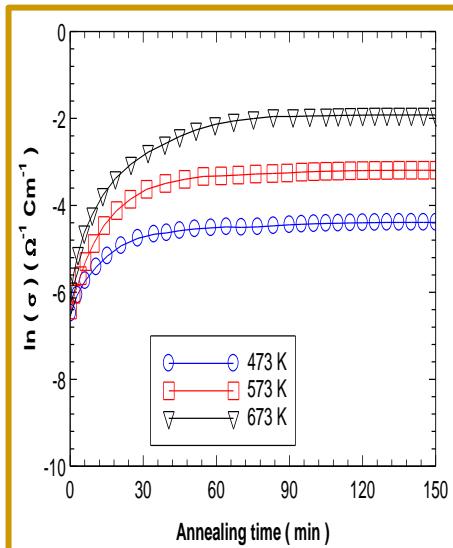


Figure.(5): logarithm.of.the.electrical.conductivity.versus.the.annealing.time.at.constant.temperatures.for.Si₁₀Ge₁₀Zn₈₀.multilayers.

IV. Conclusions

The samples show the same general behavior where the electrical conductivity increases with increasing the annealing time at constant annealing temperature and becomes constant at high annealing time. This behavior is attributed partially to crystallization occurring in the multilayers, especially for germanium atoms depending on the annealing temperature and annealing time as confirmed by X-ray dat, and due to the incorporation of zinc or

tellurium to the network. The elements of dopes at high concentration plays an important role for determining the electrical conductivity and optical gap in this study the conductivity are (ptyp e) similarly for holes. The reason in creating optical gap, this may be due to shifting the Fermi level, where increasing shifting the Fermi level towards the valence band upon zinc or tellurium doping in the thin films..

It was found the electrical conductivity increasing and the optical gap decreasing with the crystallization effects occur only in Ge matrix and with increasing zinc at high concentration. Where the optical gap change from 1.35eV to 1.22eV at high concentration for tellurium and zinc respectively. It is clear that the change of the electrical conductivity with annealing time at different isotherms is one of the sensitive physical properties to reflect the change and growth of the phase transformation process of any material

V. Acknowledgements

My great thanks are due to Dr. Mohamed Nawwar, Dr. of solid state physics, Physics Department, Faculty of Science, Menoufia University, E



gypt.for.his.kind.an.immensely.help.in.achieving.my.goals..This.work.is.sponsored.by.the.Libyan.ministry.of.higher.education.and.Benghazi.University..

References

- [1].C.N.R.Rao.and.K.J..Rao.,phase.transformation.in.solids.,91(1978)..
- [2].D.Rurnbull,,solid.state.physics.e.seitz.and.turnbull.,,Accademic.press.New.York.,3(1959).
- [3].M..Avrami,Chem..Phys.7,1103(1939).
- [4].M.K..El-Mously.and.M.M.El-Zaidia,,J..Non-cryst-solids.27,265(1978).
- [5].M.M.El-Zaidia.and.A.M.Nassar.,physics.and.chemistry.of.glasses.22,147(1981).
- [6].M.M.El-Zaidia,Ind.,J..Tech.,18,299.(1980).
- [7].Deepika,K..S..Rathore,N..S..Saxena,New.Journal.of.Glass.and.Ceramics.,2,1(2012)..
- [8].C.N.R..Rao.and.K.J..Rao.,phase.transformation.in.solids.93(1978)..
- [9].J.N..Hay.,Brit..Polym..J.,374(1971).
- [10].M..S..Abo.Ghazala,E..Aboelhasan.,A..H..Amar.,and.W..Gamel.,Phys.

Status.Solidi.C.8.,No..11–12,.3095–3098.(2011).

[11].B..Tillack,,P..Zaumseil,G..Morgenstern,D..Kruger.,B..Dietrich.,G..Ritter,Journal.of.Crystal.Growth.,Vol..157.,pp..181-184.(1995).

[12].Z..Remes.,Ph.D.Thesis.,Faculty.of.Mathematics.and.Physics.of.the.Charles.University.Institute.of.Physics.of.the.Academy.of.Sciences.of.the.Czech.Republic.,Prague.(1999).

[13].B..Thangaraju.,and.B..Kalianna n.,J.Cryst..Res.Technol..35.,71(2000)

[14].O..Stenzel,The.Physics.of.Thin.Film.Optical.Spectra.,(Springer-Verlag.,Berlin.,Heidelberg.,Germany).Press.(2005).

[15].J..M..Zahler.,,A..Fontcuberta.M orral,,M..J..Griggs.,H..A..Atwaterand .Y..J..Chabal.,Phys..Rev..B.75.,35309.(2007).

[16].K..W..Jobson.and.J.P.R..Wells, R.E..I..Schropp,N.Q..Vinh,J.I..Dijkhuis.,J.Appl..Phys..103.,13106.(2008).

[17].Mursai.,S..Amiruddin,I..Usma.T.Winata,Sukirno.,and.M..Barmawi,Asian.J..Energy.Environ.,Vol.5.,Issue 3.,pp..211-222.(2004).

[18.]Masheal.Ali.Alghamdi.,**Master of.Science.,Department.of.Physics.,**



The.University.of.King.Saud.University.(2008).

[19].J..Tauc.,in.Amorphous.and.Liquid.Semiconductors.,ed.(J..Tauc.,Plenum).159.(1974).

[20].Z..Li,.Ph.D.Thesis,.Iowa.State.University,.Ames,Iowa.(2013).

[21].C..Wang,.D..Wuu,.S..Lien,.Y..Lin,.C..Liu,.C..Hsu,.and.C..Chen,.International.Journal.of.Photoenergy,.Vol 2012,.p..6.(2012).

[22].M..S..Abo.Ghazala,.P.hys..Status.Solidi.C.8.,3099–3102.(2011).

[23].C..Miha.,F..Sava,I..D..Simand.alca,.I..Burducea,.N..Becherescu.&.A. Velea,.Scientific Reports,.11:.11755..(2021).

[24].A..A..Langford,.M..L..Fleet,.B..P..Nelson,.W..A..Lanford.and.N..Maley,.Phys..Rev..B.45,.13367.(1992).

[25].J..A..Augis,J..E..Bennett,.J..Theoretical.Anal..13,.283.(1978).

[26].M..M..EL-Zaidia,.A..Shafi,.A.A.Ammar,.M..Abd.O.Ghazala,.J..Mat..Sci..22,.1618.(1987)

[27].Pradeep,.N.S..SaxenaA..Kumar..Physica.Scripta.54,.207.(1996)